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# Preparation of  $Mn_3O_4$  nanocrystallites by low-temperature solvothermal treatment of  $\gamma$ -MnOOH nanowires

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#### Abstract

The preparation of trimanganese tetroxide  $(Mn_3O_4)$  nanocrystallites from y-MnOOH nanowires under mild conditions has been achieved by two steps: first,  $\gamma$ -MnOOH nanowires with a mean diameter of about 12 nm and lengths of up to several micrometers were directly prepared via hydrothermal reaction between  $KMnO_4$  and toluene in water at 180°C for 24 h; then, pure  $Mn_3O_4$ nanocrystallites could be obtained by solvothermal treatment of the  $\gamma$ -MnOOH nanowires in ethylenediamine (EDA) and ethylene glycol (EG) at 150°C for 24 h. It was found that the  $Mn_3O_4$  product obtained in EDA comprised well-defined nanocrystallites with the size in the range of 15–35 nm, while the one obtained in EG consisted of aggregated nanoparticles with the size of less than 18 nm. The possible formation mechanism of nanocrystalline  $Mn_3O_4$  in EDA and EG and reasons for the different effects of various solvents on the products were also proposed.

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Keywords: Manganese oxides; Nanocrystallites; Crystal growth; Solvothermal

# 1. Introduction

The powder of trimanganese tetroxide  $(Mn<sub>3</sub>O<sub>4</sub>)$  is known to be an active catalyst for the oxidation of methane and carbon monoxide [\[1\],](#page-4-0) the selective reduction of nitrobenzene [\[2\],](#page-4-0) or the combustion of organic compounds at temperatures of 373–773 K which is of interest in alleviating several air-pollution problems [\[3,4\].](#page-4-0) It has also been used as a starting material in the preparation of Li–Mn–O electrode materials for rechargeable lithium batteries [\[5,6\]](#page-4-0) and soft magnetic materials such as manganese zinc ferrite which is useful for magnetic cores in transformers for power supplies [\[7\]](#page-4-0). Nanometer-sized  $Mn_3O_4$  powders with remarkably increased surface area are expected to display better performance in all the above-mentioned aspects of applications. For instance, a much larger surface-tovolume ratio can surely improve the catalysis capacity of  $Mn_3O_4$  powders due to more adsorption of oxygen

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for the oxidation of carbon monoxide and nitrogen oxides [\[1,8,9\].](#page-4-0)

 $Mn<sub>3</sub>O<sub>4</sub>$  powders were usually prepared by heating either manganese oxides with a higher valence of manganese (e.g.,  $MnO_2$ ,  $Mn_5O_8$ , and  $Mn_2O_3$ , etc.) or Mn(II) and Mn(III) hydroxide, oxyhydroxide, carbonate, nitrate, and sulfate at about  $1000^{\circ}$ C in air [\[10–12\]](#page-4-0). However, these calcining methods were apt to yield products with a form of hard solid or coarse grains. Another conventional process to synthesize  $Mn_3O_4$ powders was sol–gel technique, which still need posttreatment at high temperatures [\[13,14\]](#page-4-0). Currently, there is obviously a trend toward devising simple, lowtemperature solution chemical synthetic methods for nanomaterials preparation. Hydrothermal/solvothermal technique is among the most promising solution chemical routes to synthesize nanocrystallites generally below  $250^{\circ}$ C [\[15,16\]](#page-4-0). The distinct advantage of such a soft solution process is to induce the preparation of high-crystallized powders with narrow grain size distribution and high purity without post-treatment at high temperatures [\[17\].](#page-4-0) In addition, the particle properties such as morphology and size can be well controlled

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during the hydrothermal/solvothermal process. Recently, a low-temperature solvothermal method to  $Mn_3O_4$  nanoparticles has been successfully developed, which was based on the reaction of  $KMnO_4$  and a great excess of  $CH_3OH$  or  $CH_3CH_2OH$  (CH<sub>3</sub>OH or  $CH<sub>3</sub>CH<sub>2</sub>OH$  served as both the reducing agent and the solvent) at  $80-170$ °C [\[18\].](#page-4-0) However, the strong oxidation capability of KMnO<sub>4</sub> and large volume of flammable and toxic  $CH<sub>3</sub>OH$  or  $CH<sub>3</sub>CH<sub>2</sub>OH$  used in this solvothermal route made it somewhat dangerous.

In this work, we report a facile novel solvothermal way to prepare  $Mn_3O_4$  nanocrystallites from  $\gamma$ -MnOOH nanowires under mild conditions. First,  $\gamma$ -MnOOH nanowires with an average diameter of about 12 nm and lengths of up to several micrometers, which were used here as the special precursor to  $Mn_3O_4$  nanocrystallites, were directly prepared through hydrothermal reaction between  $KMnO_4$  and toluene in water at 180 $^{\circ}$ C for 24 h. Then, subsequent solvothermal treatment of the as-prepared  $\gamma$ -MnOOH nanowires in ethylenediamine (EDA) and ethylene glycol (EG) at  $150^{\circ}$ C for 24 h would produce phase-pure  $Mn_3O_4$  nanocrystallites, which have been characterized by powder X-ray diffraction (XRD), Fourier transform infrared spectrum (FTIR) and wet chemical analysis, and transmission electron microscopy (TEM). The possible formation mechanism of nanocrystalline  $Mn_3O_4$  in EDA and EG and reasons for the different effects of solvents on the products were also discussed.

### 2. Experimental

# 2.1. Preparation of  $\gamma$ -MnOOH nanowires and Mn<sub>3</sub>O<sub>4</sub> nanocrystallites

All the reagents used in the experiments were of analytically pure grade without any further purification. g-MnOOH nanowires were prepared via the hydrothermal reaction between KMnO<sub>4</sub> and toluene in water at  $180^{\circ}$ C for 24 h, according to the following equation:

 $3KMnO_4+2C_6H_5-CH_3\rightarrow 3\gamma$ -MnOOH  $+2C_6H5-COOK + H_2O + KOH.$ 

In a typical procedure for the preparation of  $Mn_3O_4$ nanocrystallites,  $0.5 g$  of the as-prepared y-MnOOH nanowires was put into a Teflon-lined stainless-steel autoclave of 50 mL capacity, to which 40 mL of EDA or EG was added. The autoclave was sealed and maintained at  $150^{\circ}$ C for 24 h, then it was allowed to cool to room temperature naturally. The as-formed precipitate was filtered, washed with deionized water and ethanol for several times, and dried at room temperature in the air.

## 2.2. Characterization of the as-prepared precursor and product

Powder XRD patterns of the as-prepared precursor and products were measured at a scanning rate of  $8^{\circ}$  min<sup>-1</sup> in the 2 $\theta$  range from 10 to 70°, using a Japan Mac Science M03XHF<sup>22</sup> X-ray diffractometer (CuKa radiation,  $\lambda = 1.54056 \text{ Å}$ ; 40 KV; 40 mA). The TEM images of the as-prepared precursor and products were taken with a Philips Tecnai12 TEM at an accelerating voltage of 120.0 KV. FTIR of the obtained products in the wave number range of  $1600-400 \text{ cm}^{-1}$  were recorded on a Nicolet AVARAT360 FT-IR Spectrometer at room temperature with samples in a KBr wafer. The manganese content in the obtained products was determined with the proposed EDTA titrimetric method [\[19\].](#page-4-0)

## 3. Results and discussions

Fig. 1 shows the XRD patterns of the as-prepared precursor. All its diffraction peaks could be assigned to  $\gamma$ -MnOOH. The lattice parameters obtained by refinement of the XRD data were  $a = 8.1577 \text{ Å}, b = 5.1776 \text{ Å},$ and  $c = 5.2278$  Å, which also agreed with those of y-MnOOH reported in the JCPDS Card (File No. 41- 1379). The extremely narrow and strong diffraction peak positioned at  $2\theta = 26.2^{\circ}$  may suggest the preferential growth of  $\gamma$ -MnOOH under the hydrothermal condition. From the TEM image of the as-prepared precursor shown in [Fig. 2](#page-2-0), it can be seen that bulk quantity of  $\gamma$ -MnOOH nanowires with a relatively uniform diameter of about 12 nm and length of up to several micrometers were synthesized directly from the hydrothermal reaction between  $KMnO_4$  and toluene in water at  $180^{\circ}$ C, without the addition of any surfactant as the structure-directing template. It is well accepted



Fig. 1. XRD pattern of the as-prepared precursor.

<span id="page-2-0"></span>

Fig. 2. TEM image of the as-prepared precursor.

that utilization of nanoscaled powders as the starting material can lead to milder preparation conditions for a target product [\[20,21\]](#page-4-0). However, the previously reported hydrothermal methods to MnOOH powders with elongated morphologies always need the assistance of surfactants to direct the crystallization of the reactants in the desired products [\[22,23\].](#page-4-0) The easy preparation of g-MnOOH nanowires in our study makes it more practicable to be used as a precursor to nanocrystalline  $Mn_3O_4$ .

Fig. 3 shows the XRD patterns of the obtained products via solvothermal treatment of the  $\gamma$ -MnOOH nanowires in EDA (Fig. 3a) and EG (Fig. 3b) at  $150^{\circ}$ C for 24 h. As could be seen from Fig. 3, both products displayed the similar XRD patterns corresponding to the tetragonal cell of  $Mn_3O_4$  (Hausmannite). The lattice parameters obtained by refinement of the XRD data were  $a = 5.7589$  and  $c = 9.4662 \text{ Å}$  for the Mn<sub>3</sub>O<sub>4</sub> synthesized in EDA and  $a = 5.7641$  and  $c = 9.4482$  Å for that in EG, which were also consistent with those of Mn3O4 reported in the JCPDF card (File No. 27-734). However, the diffraction peaks of the  $Mn_3O_4$  powders obtained in EG were obviously wider and weaker than those of the  $Mn_3O_4$  powders obtained in EDA, suggesting that the  $Mn_3O_4$  powders obtained in EDA were larger and better crystallized than those obtained in EG according to the Debye–Scherrer formula [\[24\].](#page-4-0)

Fig. 4 shows the FTIR spectra of the obtained  $\text{Mn}_3\text{O}_4$ powders by solvothermal treatment of the  $\gamma$ -MnOOH nanowires in EDA (Fig. 4a) and EG (Fig. 4b) at  $150^{\circ}$ C for 24 h. It can be seen from Fig. 4 that there were no characteristic IR absorption bands of EDA and EG, indicating that both of the  $Mn<sub>3</sub>O<sub>4</sub>$  products were free of



Fig. 3. XRD patterns of the obtained products via solvothermal treatment of the  $\gamma$ -MnOOH nanowires in EDA (a) and EG (b) at 150°C for 24h.



Fig. 4. FTIR spectra of the obtained Mn<sub>3</sub>O<sub>4</sub> powders via solvothermal treatment of the  $\gamma$ -MnOOH nanowires in EDA (a) and EG (b) at 150°C for 24h.

the solvents. The FTIR spectra of both products demonstrated a notable resemblance to those of  $Mn<sub>3</sub>O<sub>4</sub>$  in the previous studies [\[25–27\]](#page-4-0). In the region from 650 to  $500 \text{ cm}^{-1}$ , two absorption peaks were observed at 638 and 531 cm<sup>-1</sup> for the Mn<sub>3</sub>O<sub>4</sub> obtained in EDA or at 642 and 521  $\text{cm}^{-1}$  for that obtained in EG. These absorption peaks may be associated with the coupling modes between the Mn–O stretching modes of tetrahedral and octahedral sites [\[27\].](#page-4-0) In the region from 500 to  $400 \text{ cm}^{-1}$ , the absorption peaks at 423 and  $417 \text{ cm}^{-1}$  were observed for the Mn<sub>3</sub>O<sub>4</sub> obtained in EDA and EG, respectively. These absorption peaks were attributed to the band stretching modes of the octahedral sites; displacement of the  $Mn^{2+}$  ions in the tetrahedral sites was negligible [\[27\]](#page-4-0). Moreover, the manganese contents in both products determined by the proposed EDTA titrimetric method [\[19\]](#page-4-0) were close to 72.0%  $(\text{m m}^{-1})$ , which corresponds to the theoretical value of manganese content in  $Mn_3O_4$ . Thus, the FTIR spectra and wet chemical analysis further confirmed that the obtained products in both solvents were composed of  $Mn_3O_4$ .

Fig. 5 shows the TEM images of the obtained  $Mn_3O_4$ powders by solvothermal treatment of the  $\gamma$ -MnOOH nanowires in EDA (Fig. 5a) and EG (Fig. 5b) at  $150^{\circ}$ C for 24 h. It was clear from Fig. 5a that the obtained  $Mn<sub>3</sub>O<sub>4</sub>$  in EDA was composed of well-defined nanocrystallites with the size ranging from 15 to 35 nm. On



Fig. 5. TEM images of the obtained  $Mn_3O_4$  powders via solvothermal treatment of the  $\gamma$ -MnOOH nanowires in EDA (a) and EG (b) at 150°C for 24 h.

the other hand, the obtained  $Mn_3O_4$  in EG as shown in Fig. 5b consisted of aggregated nanoparticles whose sizes were less than 18 nm, together with a certain amount of amorphous substance. The TEM images of the obtained  $Mn_3O_4$  powders in two solvents were in accordance with their XRD patterns shown in [Fig. 3.](#page-2-0)

For the solvothermal treatment processes, EDA and EG were selected as the solvents chiefly due to their moderate reducibility [\[28–31\].](#page-4-0) In the high-pressure and alkalescent system, EDA mainly takes the role of electron transfer, displaying moderate intensity reductive characteristic [\[28,29\].](#page-4-0) For example, single crystalline metallic bismuth nanowires have been successfully prepared by a low-temperature  $(160^{\circ}C)$  solvothermal method through the reduction of  $Bi(NO<sub>3</sub>)<sub>3</sub> \cdot 5H<sub>2</sub>O$  by EDA, which acts as a solvent as well [\[29\].](#page-4-0) In like manner, as a moderate reducing agent, EG has been widely employed in the reduction of a suitable inorganic/organic metal salt under solvothermal conditions for producing nanostructured metal powders [\[31,32\]](#page-4-0). Therefore, it is suggested that in our case, the formation of  $Mn_3O_4$  nanocrystallites by solvothermal treatment of  $\gamma$ -MnOOH nanowires in pure EDA and EG at 150 $\degree$ C may also be from a typical solvothermal reduction process. Being strongly polarizing and chelating solvents, EDA and EG can greatly enhance solubility, diffusion, and reactivity of the reactant species. Especially, when EDA and EG coordinated with  $Mn^{3+}$ , the short distance between manganese and nitrogen or oxygen atom in the coordination compound is favorable for electron to transfer from N or O to  $Mn^{3+}$  [\[29,30\]](#page-4-0), resulting in the reduction of  $\gamma$ -MnOOH to Mn<sub>3</sub>O<sub>4</sub>. Meanwhile, the elevated pressure autogenerated in the solvothermal system affords a greater driving force for the reaction process [\[32,33\].](#page-4-0) To improve our understanding of the solvothermal reduction mechanism, experiments were performed under similar conditions when EDA and EG were replaced by several nonreducibility (or less reducibility) solvents such as tetrachloride, n-hexane, benzene, pyridine, acetone, and dimethylsulfoxide. The analysis results demonstrated that in these non-reducibility (or less reducibility) solvents, the solid-state phase was still  $\gamma$ -MnOOH, indicating the solvothermal reduction reactions were not initiated. Owing to the different chemical and physical properties of EDA and EG, such as basicity, chelation, vapor pressure, and viscosity, etc. [\[34\]](#page-4-0), they exerted a diverse influence on the formation of  $Mn<sub>3</sub>O<sub>4</sub>$ . The chances are that the stronger basicity and chelation, less tendency to form dimer and trimer at high temperature, higher vapor pressure, and lower viscosity of EDA favor the solubility, diffusion, reaction of the reactant species, or the subsequent crystal growth of  $Mn<sub>3</sub>O<sub>4</sub>$ . Consequently, when the reaction temperature and reaction time were kept constant (e.g.,  $150^{\circ}$ C,  $24$ h),

<span id="page-4-0"></span>the solvothermally derived  $Mn_3O_4$  in EDA would grow larger with higher crystallinity than the one in EG. However, the detailed reaction mechanism between  $Mn_3O_4$  and EDA or EG under solvothermal conditions is still under investigation.

## 4. Conclusions

In summary,  $\gamma$ -MnOOH nanowires with an average diameter of about 12 nm and lengths of up to several micrometers were directly prepared through hydrothermal reaction between KMnO4 and toluene in water at  $180^{\circ}$ C for 24 h, and phase-pure Mn<sub>3</sub>O<sub>4</sub> nanocrystallites were successfully synthesized by solvothermal treatment of the y-MnOOH nanowires in EDA and EG at a relatively low temperature of  $150^{\circ}$ C for 24 h, which have been characterized using XRD, FTIR, TEM, and wet chemical analysis. The as-synthesized  $Mn_3O_4$  in EDA comprised well-defined nanocrystallites with the size in the range of 15–35 nm, while the one obtained in EG consisted of aggregated nanoparticles with the size of less than 18 nm. It was suggested that the formation of  $Mn_3O_4$  nanocrystallites in EDA and EG be derived from a typical solvothermal reduction process. The possible reasons for the different effects of solvents on the products were also discussed. The proposed lowtemperature solvothermal methods to  $\gamma$ -MnOOH nanowires and  $Mn<sub>3</sub>O<sub>4</sub>$  nanocrystallites in the present work were simple, safe, and low cost, which made it very suitable to be scaled up for industrial production.

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